

Synthesis and bulk polymerisation of T-shaped enynes

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Novel T-shaped enynes have been synthesised and polymerised in the bulk to produce conjugated polymers with third order non-linear susceptibility as high as 10^{-9} e.s.u.

Progress in non-linear optical (NLO) materials in the past decade has been stimulated by the development of organic materials.¹

The present authors are currently developing new types of NLO materials; among them polymers with discrete conjugated units which combine good environmental stability, solubility and film-forming properties with reasonably high $\chi^{(3)}$ values.^{2–6} Another type of NLO material are polymers, obtained by the molten-state polymerisation of aromatic diacetylenes. When polymerised in the molten state some aromatic diacetylenes produce polymers as a transparent red glass.⁷ They showed $\chi^{(3)}$ values up to 10^{-10} e.s.u.

This paper deals with the synthesis and bulk polymerisation of T-shaped derivatives of C_6H_4 -diethynylphenyl styrene bearing donor allyloxy and acceptor nitro groups.[†] As has been shown earlier, the diacetylenes bearing a C_6H_4 -diethynylphenyl styrene group have low melting points and polymerise rapidly in the molten state to produce a transparent glassy polymer with $\chi^{(3)}$ of 10^{-10} e.s.u.⁷

The key step in the synthesis of low molecular weight precursors in the Heck reaction between C_6H_4 -dibromostyrene derivatives **3** and **11** prepared from the corresponding

[†] Experimental. 4-Ethynylnitrobenzene **5**, 4-allyloxybenzaldehyde **2** and 4-allyloxyiodobenzene **8** were prepared according to the literature.^{8,10} Methylene dichloride was distilled over CaH_2 . Other reagents were used as received from Aldrich. The synthesis of the monomers is shown in Scheme 1.

Solution ¹H and ¹³C NMR spectra were recorded at 300 and 75.5 MHz, respectively, in CDCl_3 with TMS as an internal standard. Solid-state NMR spectra were recorded on a CMX-300 spectrometer at 75.5 MHz. Thermal polymerisation of monomers was carried out as follows: 0.01 g of the monomer was sandwiched between glass-plates and heated under nitrogen. The temperatures and polymerisation times are listed in Table 1. The solid dark-red glassy polymer formed between glass plates was used to measure $\chi^{(3)}$ values.

4-Allyloxybenzaldehyde **2**: ¹H NMR, δ : 9.87 (s, 1H, CHO), 7.82 (d, 2H, arom., *ortho* to CHO, *J* 8.7 Hz), 7.00 (d, 2H, arom., *meta* to CHO, *J* 8.7 Hz), 6.11–5.98 (m, 1H, $\text{OCH}_2\text{CH}=\text{CH}_2$), 5.44–5.30 (m, 2H, $\text{OCH}_2\text{CH}=\text{CH}_2$), 4.62 (td, 2H, $\text{OCH}_2\text{CH}=\text{CH}_2$, *J* 1.5 Hz, *J* 5.4 Hz); ¹³C NMR, δ : 170.3 (CHO), 163.8 ($\text{C}_{\text{ar}}-\text{O}$), 132.5 ($-\text{CH}=\text{C}$), 131.9 (C_{ar} *ortho* to CHO), 130.0 ($\text{C}_{\text{ar}}-\text{C}$ HO), 118.5 ($\text{CH}_2=\text{C}$), 115.0 (C_{ar} *meta* to CHO), 69.0 (CH_2).

4-Allyloxy-*o*,*o*-dibromostyrene **3**: ¹H NMR, δ : 7.47 (d, 2H, *ortho* to $\text{CH}=\text{CBr}_2$, *J* 8.7 Hz), 7.37 (s, 1H, $-\text{CH}=\text{CBr}_2$), 6.87 (d, 2H, *meta* to $\text{CH}=\text{CBr}_2$, *J* 8.7 Hz), 6.08–5.96 (m, 1H, $-\text{C}=\text{H}$), 5.40–5.25 (m, 2H, $=\text{CH}_2$), 4.52 (d, $-\text{OCH}_2$, *J* 5.4 Hz); ¹³C NMR, δ : 158.7 ($\text{C}_{\text{ar}}-\text{O}$), 136.3 ($-\text{CH}=\text{CBr}_2$), 132.9 ($-\text{CH}=\text{C}$), 129.8 (C_{ar} *meta* to $-\text{O}$), 127.9 ($\text{C}_{\text{ar}}-\text{CH}=\text{CBr}_2$), 117.8 ($=\text{CH}_2$), 114.6 (C_{ar} *ortho* to $-\text{O}$), 87.3 ($=\text{CBr}_2$), 68.7 (CH_2).

4-Ethynylnitrobenzene **5**: mp 152–153 °C. ¹H NMR, δ : 8.20 (d, 2H, *ortho* to NO_2 , *J* 8.7 Hz), 7.62 (d, 2H, *meta* to NO_2 , *J* 8.7 Hz), 3.35 (s, 1H, $\text{HC}\equiv\text{C}$).

4-Allyloxy-*o*,*o*-di(ethynyl-4'-nitrophenyl)styrene **6**: mp 192–193 °C, yield 32%. ¹H NMR, δ : 8.28–8.21 (m, 4H, H^{14} , H^{14}), 7.90 (d, 2H, H^5 , *J* 8.7 Hz), 7.68–7.60 (m, 4H, H^{13} , H^{13}), 7.27 (s, 1H, H^8), 6.98 (d, 2H, H^5 , *J* 8.7 Hz), 6.14–6.01 (m, 1H, H^2), 5.48–5.31 (m, 2H, H^1), 4.61 (td, 2H, *J* 1.5 Hz, *J* 5.1 Hz); ¹³C NMR, δ : 160.3 (C^4), 147.13 and 147.12 (C^{15} and C^{15}), 146.3 (C^8), 133.4 (C^2), 132.3 and 131.2 (C^{14} , C^{14} and C^6), 129.9 and 129.8 (C^{12} and C^{12}), 128.1 (C^7), 123.8 and 123.7 (C^{13} and C^{13}), 118.1 (C^1), 114.9 (C^5), 99.0, 98.9, 94.0, 92.2 and 91.6 (C^9 , C^{10} , C^{10} , C^{11} and C^{11}), 68.9 (C^3).

Table 1 Some properties of low molecular weight precursors and the respective polymers.

Compound	mp/ °C	λ_{max}^a / nm	τ_{polym}^b / min	T_{polym}^c / °C	T_{10}^d / °C	T_{max}^e / °C	$\chi^{(3)} \times 10^{-10}$ (e.s.u.)
6	192	372	3	200	310	250	30
12	107	387	5	170	280	175	16
13	200	365	2	210	240	250	—

^aLong-wave absorption maximum. ^bPolymerisation time. ^cPolymerisation temperature. ^d10% weight loss temperature (heating rate 10 °C min⁻¹). ^eExotherm maximum.

benzaldehydes **2** and **10** by the Wittig reaction, and previously synthesised substituted phenyl acetylenes **5** and **9**. The Heck reaction between compounds **10** and **9** gave monosubstituted enyne **13** in addition to the main product **12**. Compound **13** is also highly unsaturated, bears donor and acceptor groups, and was polymerised in the bulk. Most likely, the formation of a considerable amount of compound **13** in contrast to the Heck reaction between compounds **2** and **5** where no monosubstituted side product was isolated is connected with the low acidity of acetylene **9** compared to **5**, making the Heck reaction more difficult. The structure of the monomers **6**, **12** and **13** was

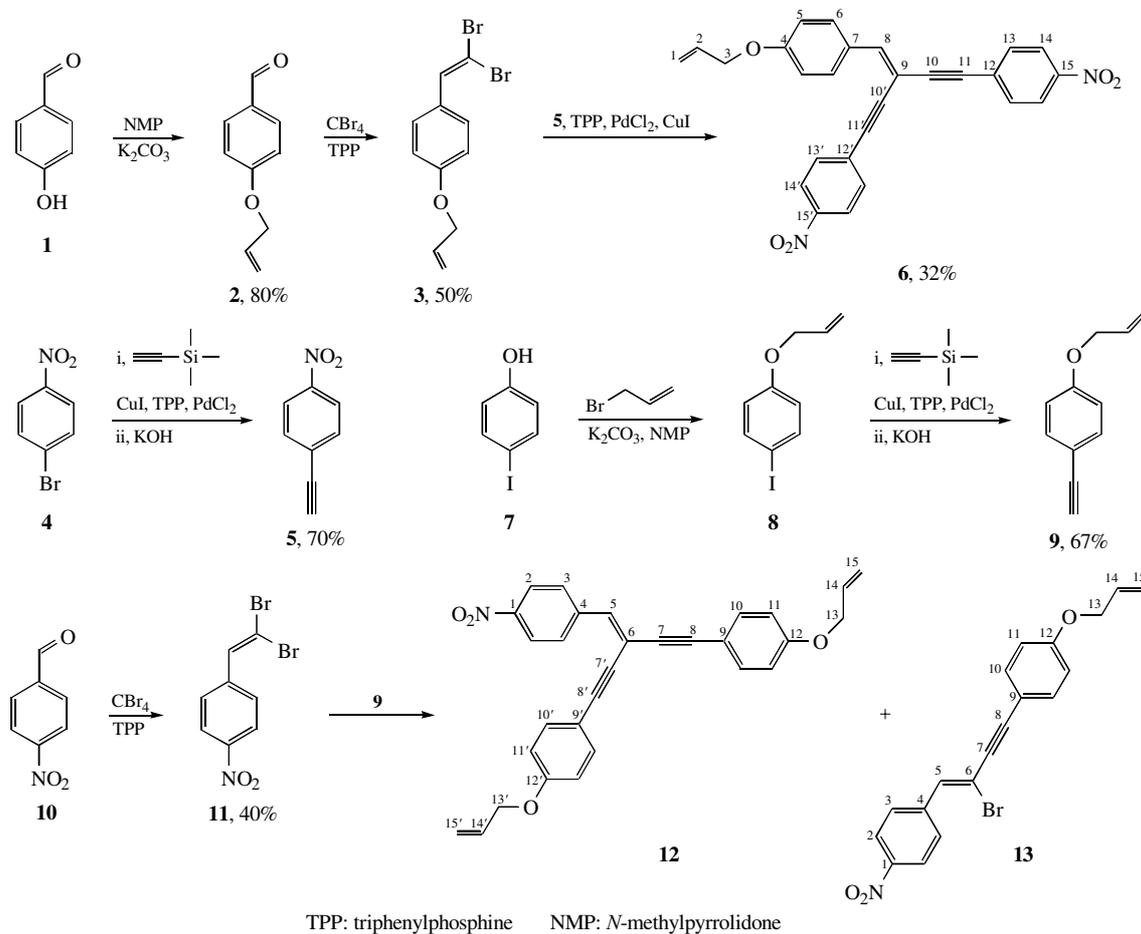
4-Allyloxyiodobenzene **8**: ¹H NMR, δ : 7.52 (d, 2H, *ortho* to I, *J* 9.0 Hz), 6.67 (d, 2H, *meta* to I, *J* 9.0 Hz), 6.07–5.94 (m, 1H, $-\text{CH}=\text{C}$), 5.41–5.25 (m, 2H, $=\text{CH}_2$), 4.48 (td, 2H, *J* 1.5 Hz, *J* 5.1 Hz).

4-Ethynylallyloxybenzene **9**: overall yield 67%. ¹H NMR, δ : 7.40 (d, 2H, *meta* to $-\text{O}-$, *J* 8.7 Hz), 6.83 (d, 2H, *ortho* to $-\text{O}-$, *J* 8.7 Hz), 6.08–5.96 (m, 1H, $-\text{CH}=\text{C}$), 5.42–5.25 (m, 2H, $=\text{CH}_2$), 4.80 (td, 2H, $-\text{C}=\text{H}$, *J* 1.5 Hz, *J* 5.1 Hz), 2.98 (s, 1H, $\equiv\text{CH}$); ¹³C NMR, δ : 159.0 ($\text{C}_{\text{ar}}-\text{O}$), 133.6 (C_{ar} *meta* to $-\text{O}-$), 132.9 ($-\text{CH}=\text{C}$), 117.9 ($=\text{CH}_2$), 114.8 (C_{ar} *ortho* to $-\text{O}-$), 114.4 ($\text{C}_{\text{ar}}-\text{C} \equiv$), 83.7 ($-\text{C} \equiv$), 75.81 ($\equiv\text{CH}$), 68.8 (CH_2).

4-Nitro-*o*,*o*-dibromostyrene **11**: mp 48–50 °C. ¹H NMR, δ : 8.20 (d, 2H, *ortho* to NO_2 , *J* 7.2 Hz), 7.68 (d, 2H, *meta* to NO_2 , *J* 7.2 Hz), 7.56 (s, 1H, $\text{CH}=\text{CBr}_2$); ¹³C NMR, δ : 147.3 ($\text{C}_{\text{ar}}-\text{N}$ O_2), 141.5 ($\text{C}_{\text{ar}}-\text{CH}=\text{CBr}_2$), 135.0 ($-\text{CH}=\text{CBr}_2$), 129.2 (C_{ar} *ortho* to NO_2), 123.8 (C_{ar} *meta* to NO_2), 94.1 ($=\text{CBr}_2$).

4-Nitro-*o*,*o*-di(ethynyl-4'-allyloxyphenyl)styrene **12** and 4-nitro-*o*,*o*-di(ethynyl-4'-allyloxyphenyl)styrene **13**: a solution of compound **11** (2.9 g, 4.4 mmol), compound **9** (3.0 g, 2.94 mmol), TPP (0.7 g), CuI (0.02 g) and $\text{PdCl}_2(\text{TPP})_2$ (0.004 g) in triethylamine (60 ml) was refluxed under nitrogen for 6 h. The solvent was evaporated *in vacuo* and the residue was rinsed with water and chromatographed on SiO_2 with CCl_4 giving compounds **12** and **13** with yields of 31% and 25%, respectively. Compound **12**: mp 106–107 °C. ¹H NMR, δ : 8.22 (d, 2H, H^2 , *J* 8.7 Hz), 6.90 (d, 2H, H^3 , *J* 8.7 Hz), 7.47 (d, 4H, H^{10} and H^{10} , *J* 8.7 Hz), 7.10 (s, 1H, H^5), 6.94–6.87 (m, 4H, H^{11} and H^{11}), 6.12–5.99 (m, 2H, H^{14} and H^{14}), 5.45–5.29 (m, 4H, H^{15} and H^{15}), 4.59–4.55 (m, 4H, H^{13} and H^{13}); ¹³C NMR, δ : 159.6 and 159.3 (C^{12} and C^{12}), 147.1 (C^1), 142.2 (C^4), 138.3 (C^5), 133.4 (C^{10} and C^{10}), 133.1 and 132.9 (C^{14} and C^{14}), 129.2 and 123.7 (C^2 and C^3), 118.1 and 118.0 (C^{15} and C^{15}), 115.2 and 115.0 (C^{11} and C^{11}), 114.7 and 114.4 (C^9 and C^9), 108.5 (C^6), 96.8, 90.7, 87.9 and 85.7 (C^7 , C^8 , C^7 and C^8), 68.9 (C^{13} and C^{13}).

Compound **13**: mp 199–200 °C. ¹H NMR, δ : 8.20 (d, 2H, H^2 , *J* 9.0 Hz), 7.63 (d, 2H, H^3 , *J* 9.0 Hz), 7.47 (d, 2H, H^{10} , *J* 9.0 Hz), 7.26 (s, 1H, H^5), 6.88 (d, 2H, H^{11} , *J* 9.0 Hz), 6.10–5.97 (m, 1H, H^{14}), 5.44–5.29 (m, 2H, H^{15}), 4.65 (td, 2H, *J* 1.5 Hz, *J* 5.1 Hz); ¹³C NMR, δ : 159.9 (C^{12}), 147.4 (C^1), 134.3 and 133.1 (C^2 , C^{10} and C^5), 132.6 (C^{14}), 123.7 (C^3), 118.1 (C^{15}), 115.1 (C^{11}), 113.2 (C^9), 84.8, 79.5 and 78.7 (C^6 , C^7 and C^8), 68.9 (C^{13}).



Scheme 1 Monomer synthesis.

confirmed by standard spectroscopic techniques FT-IR and ^1H and ^{13}C NMR.

According to DSC data the polymerisation exotherms appeared immediately after the melting transition. The decomposition temperatures of the monomers **6** and **12** are of 310 and 280 °C, respectively, while compound **13** starts to decompose at 240 °C (Table 1). The low thermal stability and high melting point of **13** are due to the weak C-Br bond and the rigid linear structure of the molecule, respectively. In addition to the widest temperature range between melting point and decomposition temperature compound **12** has the most polarizable π -electron system as followed by long-wave absorption maxima (Table 1).

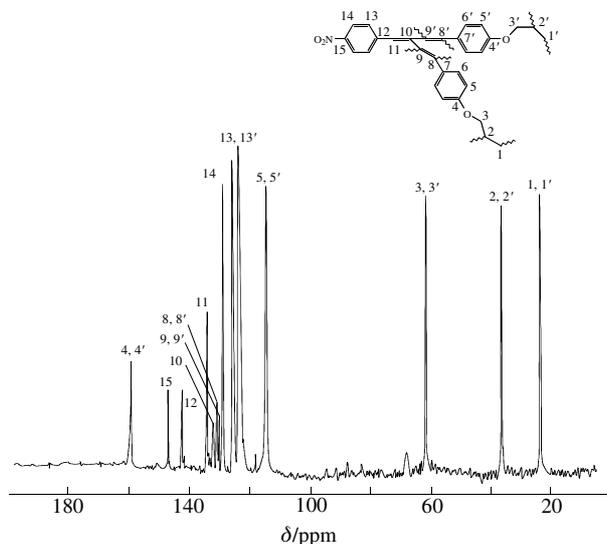
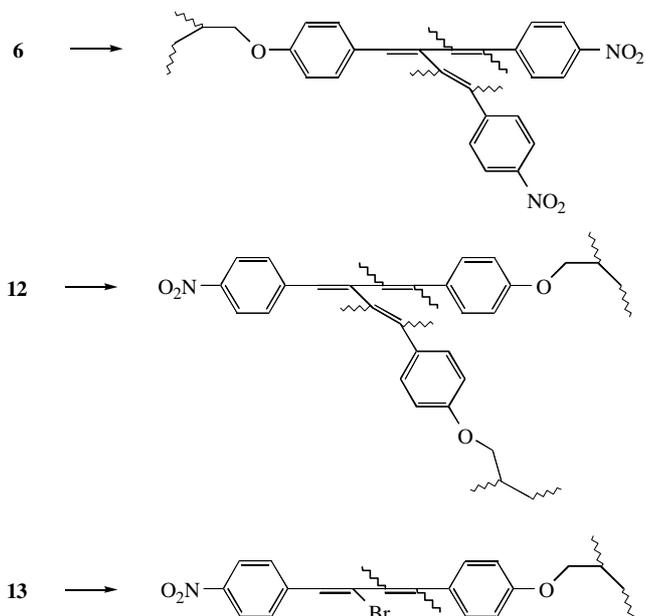


Figure 1 Solid-state NMR spectra of poly-**12** (^1H 90° pulse width 5 μs , cross-polarisation contact time 2 ms, spinning frequency 4 kHz).

All monomers polymerised in the bulk above the melting point at temperatures listed in Table 1 to produce insoluble and transparent red polymers. In the case of compound **13** the polymer obtained was not completely transparent, probably due to partial thermal composition during the polymerisation as a result of which its $\chi^{(3)}$ values could not be measured. Indeed, in the FT-IR spectra of poly-**13** a significant decrease in C-Br stretching intensity at 685 cm^{-1} compared to **13** is observed. According to solid-state ^{13}C NMR spectra polymerisation in the bulk produced polymers with a rather defined structure consisting of substituted polyacetylene sequences interconnected by polyolefin cross-links. Figure 1 shows the assigned solid-state ^{13}C NMR spectrum of poly-**12** as an example. Confirming the NMR data the FT-IR analysis shows that a very intense C \equiv C stretching at 2210 cm^{-1} in the monomers almost disappeared in the polymers. A weak peak at 1645 cm^{-1} due to the allylic C=C of **12** is missing from the spectra of poly-**12**. However, out-of-plane =CH deformation vibrations are still seen at 985 cm^{-1} . Other functional groups such as nitro (1518 and 1338 cm^{-1}) and ether (1250 and 1150 cm^{-1}) were unaffected by the polymerisation. On this basis the polymerisation route is shown in Scheme 2.

$\chi^{(3)}$ data for poly-**6** and poly-**12** obtained by using a picosecond laser consisting of a mode locked Quantel Nd:YAG laser with frequency doubled to 532 nm are listed in Table 1. It was difficult to obtain reliable data for poly-**13** as mentioned before. As seen, poly-**6** and poly-**12** show high $\chi^{(3)}$ values, $> 10^{-9}$ e.s.u. This is one order of magnitude higher than that of the similar molten-state polymerised compound bearing a C_6H_4 -diethynylphenylstyrene group.¹⁷ In all likelihood it is the polyacetylene sequences substituted by donors and acceptors that contributes to the high $\chi^{(3)}$ values of poly-**6** and poly-**12**.²



Scheme 2 Bulk polymerisation.

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